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含能聚醚黏合剂的柔性链结构设计的研究进展

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摘要: 含能聚醚黏合剂作为推进剂的骨架和基体,是促进推进剂的能量水平、力学性能、加工性能等不断改善的基础。然而,聚醚中的极性含能基团阻碍了分子链的运动,降低了高分子链的柔顺性,导致力学性能下降,制约了高能固体推进剂的发展。对含能聚醚进行共聚改性,引入柔性链段提高柔顺性是获得结构多样、性能可调聚醚的有效方法。本研究介绍了近年来含能聚醚引入不同类型的柔性链段,从主链柔性链段、侧链柔性链段两个角度展开综述,阐释其对含能聚醚黏合剂力学性能和加工性能的影响,并对含能聚醚黏合剂的柔性链结构设计研究进行了展望,为设计开发新型含能黏合剂提供方向。

关键词: 聚醚类黏合剂;高分子链;柔顺性;固体推进剂

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0 引言

高能化始终是固体推进剂研制发展的主要目标,含能聚醚黏合剂作为固体推进剂的骨架和基体,对推进剂的能量性能、力学性能、加工性能等有重要影响^[1-7]。探索新型含能黏合剂对于提高固体推进剂的能量及性能有着重要意义。新型含能黏合剂体系研究的基本思路是在聚合物侧链上引入如叠氮基、硝基、硝酸酯基、二氟氨基等含能基团以提升能量水平^[8-10]。然而,由于含能聚醚侧链中的刚性基团体积大、极性大,使得高分子链单键的内旋转困难,分子链刚性增加,柔顺性降低,导致其力学性能尤其是低温力学性能下降。因此,研究人员尝试通过共聚改性的方法在高分子链中引入柔性链段,打破聚合物链结构规整性,从而降低玻璃化转变温度(T_g)和结晶温度,实现黏合剂能量水平和力学性能的同步提高^[11-16]。

在推进剂用黏合剂的早期合成研究中,以阳离子

开环聚合形成的无规共聚物为主,四氢呋喃(THF)和乙二醇(EG)作为柔性单体,常用于提高含能聚醚的柔顺性^[17-20]。3,3-双叠氮甲基环氧丁烷-四氢呋喃(BAMO-THF)无规共聚物具有优异的燃烧性能和低温力学性能,但存在固化效果差、难以回收、安全性低等问题^[21]。与无规共聚物相比,多嵌段共聚物具有较高的温度黏度敏感性、微相分离、低易损性等特点,在高能热塑性弹性体(ETPE)中应用较多^[22-23]。随着共聚醚的研究深入,出现了共聚方法单一、柔性单元种类受限、序列不可控等问题,研究人员逐渐认识到含能聚合物结构与性能的关系,开始从以下几个方面优化调整分子链结构:(1)由于聚合物分子链骨架结构的修饰与调控对于开发新型含能聚合物至关重要,通过调整嵌段共聚物的序列结构、调节嵌段的种类和长度以期拥有更好的理化性能^[24-25];(2)由于线性共聚中柔性链段增加会降低含能基团的含量,优化能力有限,从二维角度接枝柔性侧链,可以在一定程度上摆脱分子线性结构的限制,提升性能^[26-29]。

随着固体火箭领域对推进剂综合性能要求的不断提高,含能黏合剂的设计与研发已成为目前国内外的研究热点,将柔性链段引入高分子链是固体推进剂用含能聚醚黏合剂领域发展的方向之一。为此,本研究综述了近年来引入的不同类型柔性链段,从主链柔

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链段、侧链柔性链段两个角度阐述柔性链段对含能聚醚性能的影响,为探索具有优良力学性能和加工性能的新型含能黏合剂提供研究思路和支撑。

1 主链柔性链段

1.1 无规共聚

20世纪前后,为提高聚醚力学性能,人们采用共聚的方法引入柔性链段并进行了大量研究。在推进剂用黏合剂的早期合成中,阳离子开环聚合制备共聚醚是常用的共聚方法,含能单体与THF无规共聚物的合成路径如图1所示^[21,30-32]。其中,THF和EG作为柔性单体提高聚醚的柔顺性具有优异的效果。

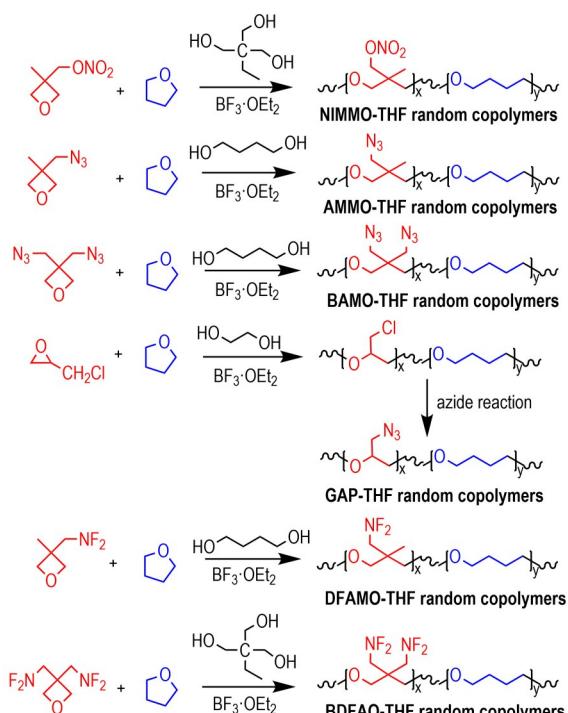


图1 含能单体与THF无规共聚物的合成路径^[21,30-32]

Fig. 1 Synthesis route for typical random copolymers from energetic monomer and THF^[21,30-32]

在热固性含能黏合剂领域,针对含能均聚物力学性能不足的问题,通过刚-柔性分子链无规排列的线性共聚实现内增塑具有广泛的应用^[33-39]。含能黏合剂共聚的初步研究主要以Manser等研究人员为主。Manser等^[17-21]陆续通过研究发现,不论是溶液聚合还是本体聚合都可以实现BAMO-THF无规共聚物的合成,不同于3,3-双叠氮甲基环氧丁烷均聚物(PBAMO),该共聚物在室温下可呈液态流动。引入柔性链节单元—(CH₂)₄O—可以打破链结构的规律性,提高链承载

系数,从而降低聚合物的T_g,增加聚合物的分子量,提高其力学性能^[40-41]。莫洪昌等^[30]以1,4-丁二醇(BDO)为引发剂,以三氟化硼乙醚为催化剂,通过3-叠氮甲基-3-甲基氧杂环丁烷(AMMO)和THF共聚将柔性链节单元—(CH₂)₄O—引入到3-叠氮甲基-3-甲基氧杂环丁烷均聚物(PAMMO)分子链中,得到3-叠氮甲基-3-甲基氧杂环丁烷与四氢呋喃共聚醚(PAT),T_g为-59.2 °C,热分解峰温为264.1 °C,THF的引入较大程度上削弱了分子内及分子间的作用力,实现了提高PAMMO低温力学性能的目的,且PAT具有良好的热稳定性。王晓川等^[42]以三羟甲基丙烷(TMP)为引发剂,三氟化硼乙醚为催化剂,将THF和3-硝酸酯甲基-3-甲基氧杂环丁烷(NIMMO)开环聚合,合成了三官能度的3-硝酸酯甲基-3-甲基氧杂环丁烷与四氢呋喃(NIMMO-THF)无规共聚醚,黏度更低,T_g为-53.7 °C,较3-硝酸酯甲基-3-甲基氧杂环丁烷均聚物(PNIMMO)(T_g=-42.4 °C)有了明显降低,热稳定性良好。

聚叠氮缩水甘油醚(GAP)作为一种高能黏合剂,因其具有正的生成焓(117.2 kJ·mol⁻¹)、高密度(1.3 g·cm⁻³)、对冲击的低敏感性及与增塑剂的相容性好而被广泛应用,但刚性叠氮基的存在限制了GAP黏合剂的流动性,力学性能更差,将GAP与EG、THF等柔顺性好的单体共聚可以获得很好的增塑效果^[43-45]。Mohan等^[31]利用三氟化硼乙醚和EG作为催化剂和助催化剂,环氧氯丙烷(ECH)和THF共聚合成了聚叠氮缩水甘油醚与四氢呋喃(GAP-THF)无规共聚物,合成的共聚物与GAP的热分解行为相似,但T_g非常低,仅为-64 °C。

此外,为解决3-二氟氨基甲基-3-甲基氧杂环丁烷(DFAMO)和3,3-偕二氟氨基甲基氧杂环丁烷(BDFAO)低温力学性能较差、不能满足实际使用要求这一问题^[46],Li等^[6,32,47-48]合成了一系列共聚物(BDFAO-THF、DFAMO-THF、DFAMO-BAMO、DFAMO-AMMO、DFAMO-NIMMO),并研究了它们的热解过程、物理和化学性质。结果表明,BDFAO-THF共聚物的T_g为-40.0 °C,较PBDFAO(T_g=130.78 °C)降低了170.78 °C,改性效果明显,DFAMO-THF共聚物的T_g为-50.8 °C,较PDFAMO(T_g=-40.8 °C)降低了10 °C,与BDFAO-THF和DFAMO-THF共聚物相比,BAMO、AMMO、NIMMO链段对二氟氨基聚合物的柔韧性没有明显提升,虽然T_g没有得到降低,但二氟氨基的存在显著提高了聚合物的整体能量水平。含THF多嵌段无规共聚醚的玻璃化转变温度和力学性能如表1所示。

表1 含THF多嵌段无规共聚醚的玻璃化转变温度和力学性能

Table 1 Glass transition temperatures of THF-containing multiblock random copolymers

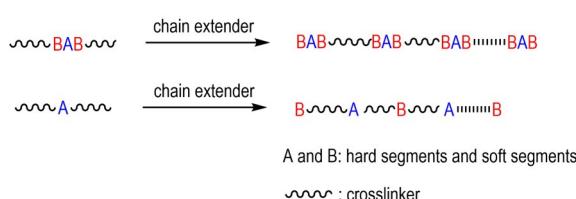
samples	molecular weight and distribution		T_g / °C	tensile strength / MPa	elongation / %
	M_n / g·mol ⁻¹	D			
AMMO-THF ^[30]	3550	1.31	-59.2	-	-
BAMO-THF ^[21]	-	-	-61.1	-	-
NIMMO-THF ^[42]	-	-	-53.7	1.32	100
GAP-THF ^[31]	1380	2.24	-64	-	-
DFAMO-THF ^[32]	2700	-	-50	-	-
BDFAO-THF ^[32]	1700	-	-40	-	-

Note: M_n is the number average molecular weight. D is the molecular weight distribution. T_g is the glass transition temperature.

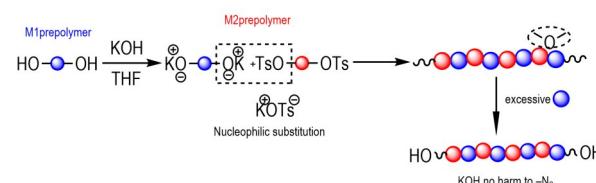
目前,在引发剂和催化剂存在的条件下,将含能单体与THF、EG柔性单体进行阳离子开环共聚,是合成无规共聚醚的主要方法。合成无规共聚物的 T_g 较均聚物明显降低,柔韧性增加,内增塑效果显著。

1.2 多嵌段共聚

由于无规共聚的不确定性,很难明确分子骨架中的序列排列。为了确定序列的调节效果,Zhai等^[20]对BAMO-THF无规共聚醚的分子链结构进行分析,发现通过控制单体的摩尔比可以获得理想的分子链结构,从而提高聚合物的力学性能。然而,调节单体摩尔比对序列结构的影响有限。和无规共聚相比,多嵌段共聚物可以通过扩链剂来调控序列结构实现对其性能的控制,从而拥有更优异的理化性能^[22-23,39]。此外,Luo等^[13-14]制备了交替BAMO-AMMO和无规BAMO-AMMO共聚物,并通过链延伸反应制备了高能热塑性弹性体ETPE,发现相对于无序排列的嵌段共聚物,交替排列的嵌段共聚物表现出更好的拉伸强度和力学性能,综合性能更好。目前,交替共聚的热塑性弹性体主要通过两种合成方法制备(图2)^[13,49]。一种方法是通过三嵌段聚合物作为黏合剂制备有序结构,另一种方法是将扩链剂与均相聚合物结合,并进行进一步的交替反应。这两种方法都需要通过加入固化剂实现扩链反应,但固化剂过多会破坏分子链的柔韧性,难以合成具有更细的交替序列的聚合物。

图2 通过扩链剂制备交替多嵌段共聚醚的两种方法^[13,49]Fig.2 Two methods to prepare alternating multiblock ethers via chain extending reactions^[13,49]

Li等^[24-25]考虑到序列不可控的问题,提出了一种不需要加入固化剂的简单方法合成序列可控的含能聚合物(ESCPs),通过亲核取代得到了链段长度可控且交替分布的嵌段共聚醚P(BAMO-alt-EG)和P(BAMO-alt-THF)。他们还合成了具有交替嵌段结构的3-叠氮甲基-3-甲基氧杂环丁烷-乙二醇共聚醚P(AMMO-ab-EG)并对其进行了表征。研究这些共聚醚发现,控制序列结构,能够有效调节其黏度和 T_g 。通过点击反应的方法引入不同的嵌段和调节序列结构,能够在不影响推进剂能量水平的情况下调节链段长度,提高柔顺性,在设计和控制推进剂用黏合剂的力学性能和热性能方面具有一定的应用潜力(图3)。

图3 BAMO基ESCPs的制备^[25]Fig.3 Synthesis route of BAMO-based ESCPs^[25]

综合来看,通过控制单体的摩尔比,能够合成理想的多嵌段共聚物,但这种方法的调节能力存在一定的局限性。为了更好地控制共聚物的性能,可以通过使用扩链剂来调整嵌段共聚物的序列结构。交替排列的嵌段共聚物表现出更好的拉伸强度和力学性能,具有较大的应用潜力。此外,采用点击反应的方法,引入不同的嵌段并调整序列结构,这也是提高聚醚黏合剂力学性能和热性能的有效策略。

1.3 三嵌段共聚

大分子引发共聚形成三嵌段共聚物大多应用于热塑性黏合剂,具有较高的温度黏度敏感性和较好低损性能,并存在微相分离现象。可以通过调节嵌段的种

类和长度保留不同链段特有的物理化学性质,提高聚合物的综合性能。同时,大分子引发剂打破了原有聚合物的规整度,是降低聚合物玻璃化转变温度的有效方法^[50-51]。

众所周知,内酯的极性大于醚的极性,这可以赋予结构更好的力学性能^[52]。PCL是一种典型的含内酯聚合物,能够通过开环共聚合成。Jutiter等^[53]研究了BCMO/ ϵ -己内酯(CL)共聚,通过无规加料发现共聚产物为AB型嵌段共聚物。Abrishami等^[57]以硝酸缩水甘油酯(GLYN)和聚己内酯(PCL)为宏观引发剂,通过开环共聚合成了三嵌段共聚物PGN-PCL-PGN($M_w=2900\text{ g}\cdot\text{mol}^{-1}$, $T_g=-56.2\text{ }^\circ\text{C}$), Bayat等^[54]合成了三嵌段共聚物PCL-GAP-PCL($M_n=1794\text{ g}\cdot\text{mol}^{-1}$, $T_g=-64\text{ }^\circ\text{C}$)。DSC结果表明,考虑到PGN($T_g=-35\text{ }^\circ\text{C}$), GAP($T_g=-48\text{ }^\circ\text{C}$),三嵌段共聚物的玻璃化转变温度大大降低。在2012年,Yang等^[55]就通过原子转移自由基聚合(ATRP)合成了聚缩水水甘油醚-b-聚叠氮乙基甲基丙烯酸酯的含能嵌段共聚物(GAP-*b*-PAEMA),并对其进行了表征(图4)。2013年Kanti等^[51]的综述中追溯了更早期的基于内酯的聚合物。

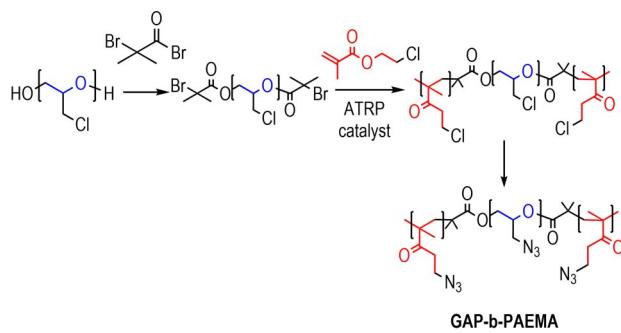


图4 应用ATRP制备的三嵌段共聚醚^[55]

Fig.4 Synthesis route of triblock copolyethers via ATRP^[55]

王晓川等^[56]以NIMMO为第一单体,烯丙基缩水甘油醚(AGE)为第二单体,通过阳离子开环聚合合成三嵌段烯基聚NIMMO含能黏合剂(图5)。黏度取决于分子间的内摩擦力,对于三嵌段烯基聚NIMMO黏合剂来说,其内摩擦力由分子间的吸引力决定。结果表明,该含能黏合剂25℃时黏度为4.76 Pa·s,具有较低的黏度^[56-57]。

Wang等^[50,58]以HTPB为引发剂,NIMMO和PGN为单体制备了新型含能黏合剂PNIMMO-HTPB-PNIMMO和PGN-HTPB-PGN。PNIMMO-HTPB-PNIMMO的 T_g 为-76℃远低于PNIMMO($T_g=-30\text{ }^\circ\text{C}$),固化后力学性能优于PNIMMO,三嵌段共聚物具有单一的玻璃

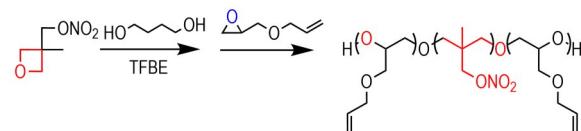


图5 三嵌段烯基聚NIMMO含能黏合剂的合成^[56]

Fig.5 Synthesis route of triblock alkene-based poly-NIMMO energetic binders^[56]

化转变温度,表明PNIMMO嵌段和HTPB嵌段存在相容性。PGN-HTPB-PGN的 T_g 为-27℃,由于其游离羟基可以在末端共聚物链上官能团化,从而提高了其力学性能。

Miriam Cappello等^[59]以HTPB作为引发剂,制备了GAP-HTPB-GAP、两种不同前体合成的GAP/BAMO-HTPB-GAP/BAMO、AMMO-HTPB-AMMO等4种不同的三嵌段共聚物,该聚合物结合了HTPB与叠氮聚醚的优点,虽然HTPB嵌段的存在使叠氮化步骤复杂化并且降低了叠氮聚合物的储存稳定性,但是可以通过改变含能嵌段的类型和长度来改善黏合剂的性能(图6)。

与传统的三嵌段共聚和扩链剂连接的多嵌段共聚不同,端羟基聚醚(HTPE)预聚物是与醚键连接的多段结构,具有优异的低温力学性能和不敏感特性, T_g 低,符合不敏感炸药(IMs)的设计要求^[60-62]。Zhang等^[5]以HTPE为引发剂,NIMMO为单体制备了新型含能黏合剂PNIMMO-HTPE-PNIMMO(图6)。PNIMMO-HTPE-PNIMMO的测试结果表明由于侧基- ONO_2 的存在增强了分子间的相互作用,低温

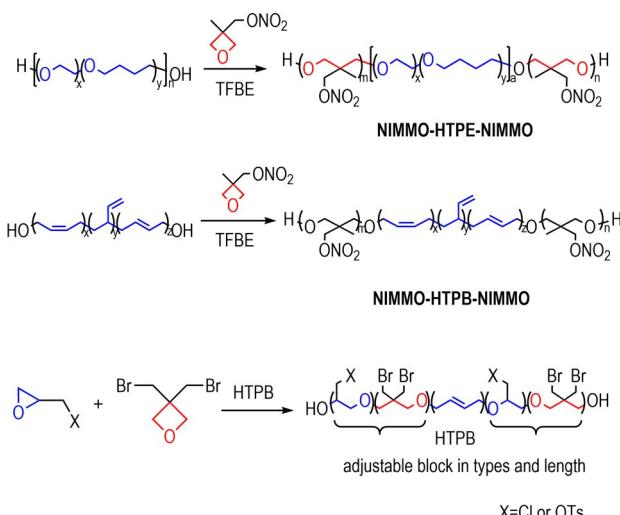


图6 以HTPB、HTPE为引发剂的三嵌段共聚醚及其合成路线^[5,50,59]

Fig.6 The synthesis routes of triblock copolyethers with HTPB as initiator^[5,50,59]

力学性能有所提高,三嵌段共聚醚的黏度比母体聚合物HTPE更高。同时,由于两种链段的极性不同导致它们的相容性变差,会进一步产生微相分离现象。

三嵌段共聚物多通过大分子引发与含能单体共聚的方法制备。引入大分子柔性链可以打破原有聚合物的规整排列,从而有效降低玻璃化转变温度。此外,通过调节嵌段的类型和长度,可以优化嵌段共聚物的综合性能。

综上所述,采用无规共聚、多嵌段共聚、三嵌段共聚等线性共聚的方法引入柔性较好的链段,能有效打破高分子链的规律性,增强柔顺性,提高含能聚醚黏合剂的力学性能。但无规共聚物具有固化效果差、难以回收、安全性低等问题,而多嵌段共聚物的温度黏度敏感性高、能够发生微相分离、易损性低,可应用性较强,同时,交替排列的嵌段共聚物表现出更好的综合性能。利用大分子作为引发剂,与含能单体共聚合成三嵌段共聚物也是降低聚合物 T_g 的有效方法。未来可通过对嵌段共聚物的序列结构、嵌段的长度和种类进行调控,开展嵌段共聚物的合成与应用的深入研究,获得性能优异的含能聚醚黏合剂。

2 侧链柔性链段

在线性增塑中,柔性链段的增加严重降低了含能基团的含量,对力学性能的优化有限,人们常通过加入增塑剂进一步提高力学性能^[63~65]。考虑到增塑剂的迁移问题^[11~13],以原有的线性骨架为基础,将柔性侧链引入含能聚合物以提高其分子间力^[27~29]。

2.1 C—H类侧链

侧链柔性链段的研究工作大部分围绕C—H类的柔性侧链展开。与PEG的对称结构相比,聚环氧丙烷(PPG)具有更好的分子柔性。由于支链甲基的存在,带来分子间和分子内作用力的优化,PPG具有优异的低温力学性能($M_n=2000\text{ g}\cdot\text{mol}^{-1}$, $T_g=-70\text{ }^\circ\text{C}$)。

2018年,Hafner等^[26]采用阳离子开环聚合法合成了ECH与1,2-环氧己烷(EpH)的线性共聚物,将非极性烷基侧链引入到GAP极性聚醚主链中,有效降低了分子间和分子内作用力,黏度与 T_g 更低。在此之后,Bayat等^[66~69]制备了线性含能三嵌段共聚物PPG-GAP-PPG($T_g=-63\text{ }^\circ\text{C}$)、PPG-PGN-PPG($T_g=-58\text{ }^\circ\text{C}$)

和无规共聚物P(GN-ran-PO)($T_g=-47\text{ }^\circ\text{C}$)。经分析,PPG-GAP-PPG和P(GN-ran-PO)均具有优良的热稳定性,且 T_g 明显低于GAP($T_g=-53\text{ }^\circ\text{C}$)和PGN($T_g=-35\text{ }^\circ\text{C}$)。此外,PPG-PGN-PPG和P(GN-ran-PO)还具有更好的力学性能。与PPG-GAP-PPG对比,PEpH-GAP-PEpH的 T_g 没有明显的优势,但考虑到分子量、序列排布及氮含量的不同,难以进行直观的比较(图7)。

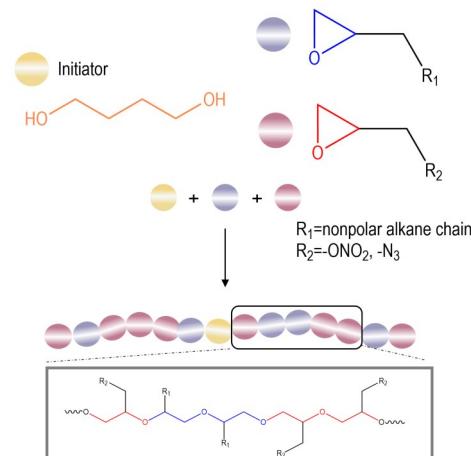


图7 非极性烷烃侧链含能共聚物的合成^[26]

Fig.7 Synthesis route of nonpolar alkane side-chain energetic copolymers^[26]

如图8所示,Golding等^[70]用3-叠氮己烷合成一种分子链更长的聚合物,称为PZ-23和PZ-24($n=6$)。PZ-23的氮含量与GAP相似, T_g 为 $-73\text{ }^\circ\text{C}$,但其能量密度比GAP高出20%以上。PZ-24是一种黏性黄色聚合物,由于碳链较长,其能量特性低于PZ-23,但因其较长的侧链提高了分子间力,其 T_g 低至 $-100\text{ }^\circ\text{C}$ 。

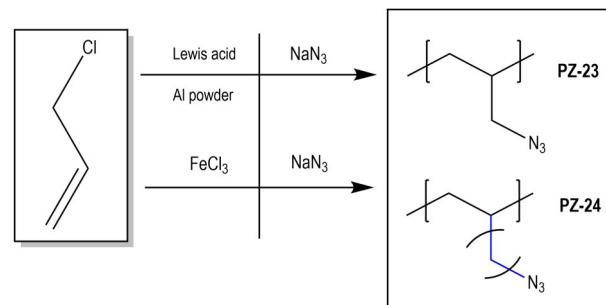
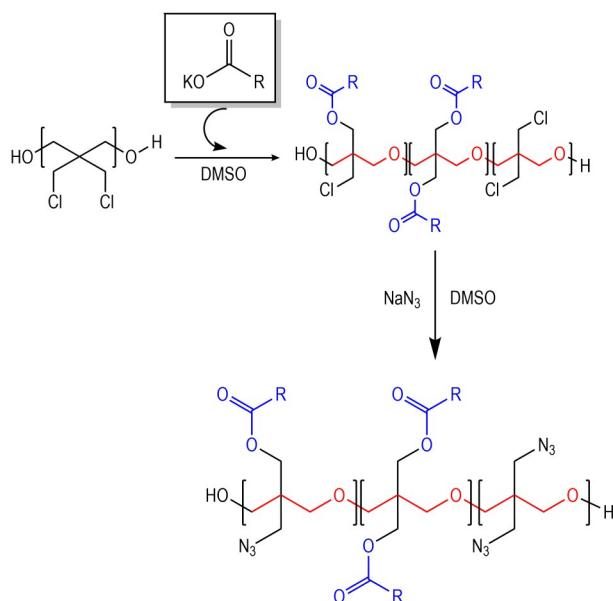


图8 PZ-23和PZ-24的合成路线^[70]

Fig.8 Synthetic routes of PZ-23 and PZ-24^[70]

Lim等^[27,71]分别在GAP和PBAMO中引入长链羧酸盐,先后制备了羧基化GAP和羧基化PBAMO(图9),羧基化GAP($T_g=-55\text{ }^\circ\text{C} \sim -48\text{ }^\circ\text{C}$)和羧基化PBAMO($T_g=-51\text{ }^\circ\text{C} \sim -43\text{ }^\circ\text{C}$)的 T_g 值均低于GAP和PBAMO,具有优越的加工性能。

图9 侧链羧基化PBAMO的合成路线^[27,71]Fig.9 Synthetic route of side-chain carboxylated PBAMO^[27,71]

目前,含C—H类侧链共聚物的合成多通过单体接枝的方法实现。C—H类侧链共聚醚的玻璃化转变温度和力学性能如表2所示。选择含有支链的柔性单体与含能单体进行共聚,合成带有侧链的共聚物,共聚物 T_g 和黏度明显降低,提高柔顺性的效果显著。

2.2 C—F类侧链

含氟类材料密度高、热稳定性、化学稳定性等均比较优异,且氟原子有助于改善炸药体系氧平衡,被认为是解决推进剂燃烧过程中产生的团聚现象的有效途径^[72-73]。由于含氟聚合物具有高密度、安全性、宽工作温度范围等优势,被认为是高爆炸性黏合剂的理想候选材料^[74-75]。为了提高GAP基聚氨酯网络黏合剂

的力学性能,Xu等^[28]通过ECH和2,2,2-三氟乙氧基甲基环氧-r-叠氮缩水甘油酯阳离子共聚,然后叠氮化,合成了一种新型氟化叠氮缩水甘油酯聚合物(聚(TFEE-r-GAP))。与GAP相比,聚(TFEE-r-GAP)基共聚聚氨酯网络具有优异的力学性能,拉伸强度为5.52 MPa,断裂伸长率为162.8%,这是一个很好的折中方案。

随后他们^[49,76]先后报道了聚[3,3-双(2,2,2-三氟-乙氧基甲基)氧杂环丁烷]乙二醇嵌段-聚(3-硝基甲基-3-甲氧基杂环丁烷)(PBFMO-b-PNMMO)和一种新型含氟共聚物P(FPO/NIMMO)。与PNMMO(6.18 MPa,635%)相比,PBFMO-b-PNMMO的拉伸强度为10.54 MPa,断裂伸长率为723%,P(FPO/NIMMO)的 T_g 为-42.7 °C较PNIMMO更低,结果表明,它们都具有优越的力学性能和加工性能。

含氟侧链共聚醚的合成路线如图10所示,他们的玻璃化转变温度和力学性能如表3所示。含C—F类侧链共聚物主要是通过含氟单体与含能单体在催化剂和引发剂存在的条件下进行阳离子开环共聚制备。由于共聚物中含有氟元素,表现出优良的力学性能和加工性能。

2.3 三唑类侧链

在分子骨架结构中接枝柔性侧链能够同时实现高能与增塑,具有一定发展潜力。但柔性侧链的长度、含量和连接位点对分子链柔顺性的影响仍然有待研究。三唑类侧链一般是通过1,3-偶极环加成反应的点击反应将反应性增塑剂接枝到聚合物上,这为接枝提供了更多的选择性。2019年,Asghar Bodaghi^[77]回顾了一系列新的反应性增塑剂,它们大多是炔烃化合

表2 C—H类侧链共聚醚的玻璃化转变温度和力学性能

Table 2 Glass transition temperatures of C—H side chain copolymers

samples	molecular weight and distribution		T_g / °C	tensile strength / MPa	elongation / %
	M_n / g·mol ⁻¹	D			
PPG-GAP-PPG ^[66-67]	1639	1.50	-63	-	-
PPG-PGN-PPG ^[68]	1886	2.02	-58	-	-
P(GN-ran-PO) ^[69]	1152	1.27	-47	0.59	109
PEpH-GAP-PEpH ^[26]	1016~1856	1.33~2.25	-60~-47	-	-
PZ-23 ^[70]	-	-	-73	-	-
PZ-24 ^[70]	-	-	-100	-	-
GAP ^[71]	1853	1.16	-49	-	-
Carboxylated GAP ^[71]	2257~2611	1.15~1.17	-55~-48	-	-
PBAMO ^[27]	2030	1.21	-37	-	-
Carboxylated-BAMO ^[27]	2080~2440	1.22~1.3	-51~-43	-	-

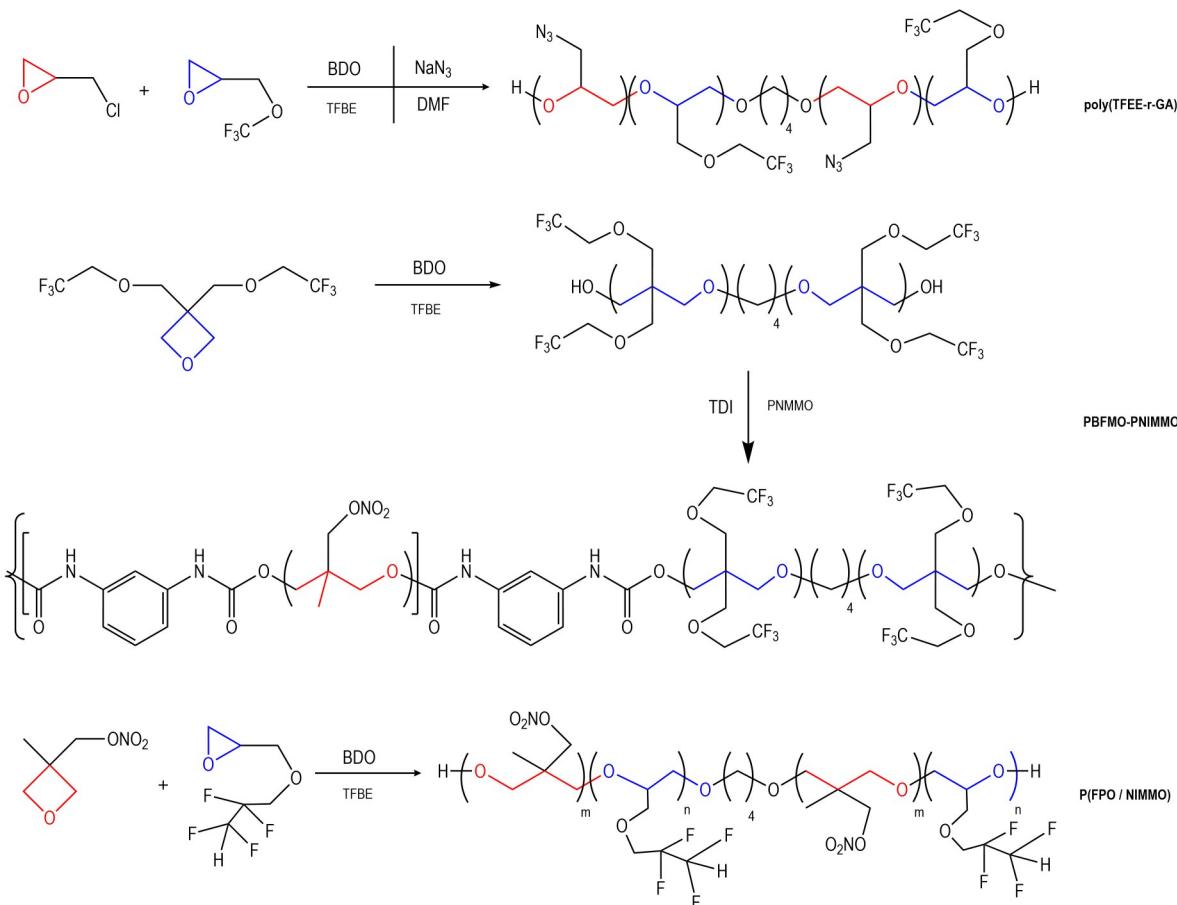
图 10 含氟侧链含能共聚物的合成^[28,49,76]Fig.10 Synthesis route of fluorinated side-chain energetic copolymers^[28,49,76]

表 3 C—F类侧链共聚醚的玻璃化转变温度和力学性能

Table 3 Glass transition temperatures and mechanical properties of C—F side chain copolyethers

samples	molecular weight and distribution		T_g / °C	tensile strength / MPa	elongation / %
	M_n / g·mol ⁻¹	D			
(poly(TFEE-r-GA)) ^[28]	2876	1.47	-49.5	5.52	162.8
P (FPO/NIMMO) ^[76]	2970	-	-42.7	-	-
PNMMO ^[76]	2610	-	-35.6	6.18	635
(PBFMO-PNMMO) ^[49]	6840	1.22	-20.4	10.54	723

物,通过叠氮化物和乙炔基团之间的点击反应与黏合剂的侧链相连(图 11)。

2020 年,Kwon 等^[78]将两种降冰片烷基反应性增塑剂(NRP)连接到聚叠氮缩水甘油四氢呋喃(PGT)基聚氨酯(PUs)上,提供了可靠且自稳定的增塑剂,同时通过防止迁移减轻了对环境的危害。NRP 的含量随着三唑基团数量的增加而增加,观察到的拉伸强度增强可能归因于新形成的三唑基团和庞大的降冰片烷部分,这些部分属于具有空间位阻效应物种。此外,将含炔烃的反应性单环增塑剂(RMCPs)化学掺入叠氮缩水甘油聚醚(GAP),以减少固化聚合物黏合剂的不稳

定性^[79]。RMCP 对 GAP 预聚物表现出的温和且独特的无催化剂点击反应可用于定制特定材料的适当加工性配方。实验测定的点击反应活性与前沿分子轨道的理论预测结果一致。将 RMCP 添加到 GAP 基聚氨酯(PU)黏合剂中,由于叠氮化物转化为三唑基团,黏合剂的拉伸性能和抗冲击性得到增强。

三唑类侧链主要是通过 1,3-偶极环加成反应的点击反应将反应性增塑剂接枝到聚合物上,这种方法制备的黏合剂的拉伸性能和抗冲击性得到增强,力学性能更好。

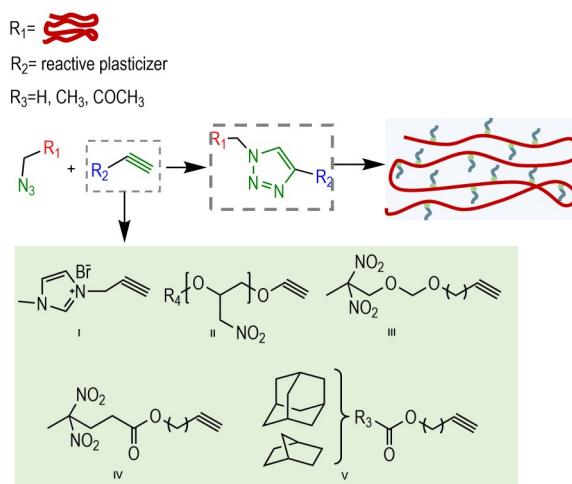
图 11 通过 1,3-偶极环反应连接的反应性含能增塑剂^[77]

Fig.11 Reactive energetic plasticizers linked via 1,3-dipole ring reactions^[77]

2.4 超支化类侧链

不同于常见的侧链型柔性链段,三维(3D)结构的超支化聚合物的主链和支链难以分辨,这类聚合物具有大量的官能团、较低的溶液或熔体黏度、无或较低的链纠缠度和较好的溶解度^[80-81]。1990年,支化GAP就被提出并合成,支化GAP比线性GAP燃烧产生更多的能量,黏度更低,玻璃化转变温度也有所降低,有望提高推进剂的力学性能^[82-83]。鉴于这些优点,近年来研究人员初步合成了一些三维文化结构的高能聚合物并对其进行了一系列表征。

超支化聚合物由于其高文化密度、较小的原子空间排列尺寸、多个锚定点和较低的摩尔纠缠度,表现出较弱的分子间相互作用和更高的顺应性。此外,它们在强制挤压或轧制过程中容易变形,这可能提高了材料的流变特性^[84-85]。超支化聚合物通常有两种结构:树枝状聚合物(图12a)和星形聚合物(图12b)。Nair等^[86]利用后改性方法将第一代和第二代树枝状分子的末端羟基转化为叠氮化基,得到端叠氮化的树枝状聚酯。然而,在端基修饰后,树枝状大分子没有活性羟基。因此,它只能作为增塑剂,而不是黏合剂。

王晓等^[87-88]以端磺酰氧基超支化聚酯为前驱体,在二甲基亚砜中与叠氮化钠反应,端磺酰氧基被叠氮基所取代,制备得到端叠氮基超支化聚酯。在此基础上,对端叠氮基超支化聚酯的合成反应进行了优化,结果表明,与端叠氮基树形聚酯相比,端叠氮基超支化聚酯的玻璃化转变温度更低,在热分解等性能方面优于前者。

Zhang等^[84]通过阳离子开环聚合合成了一系列新

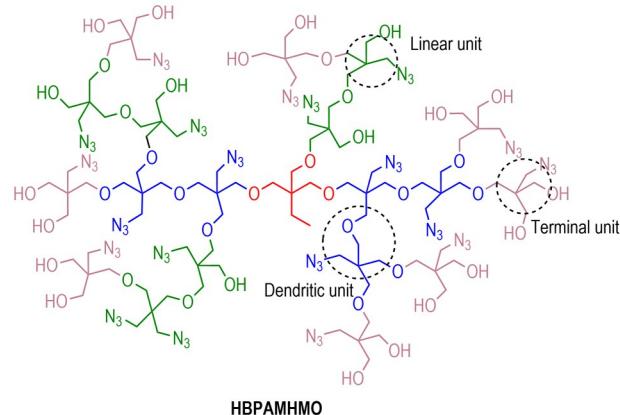
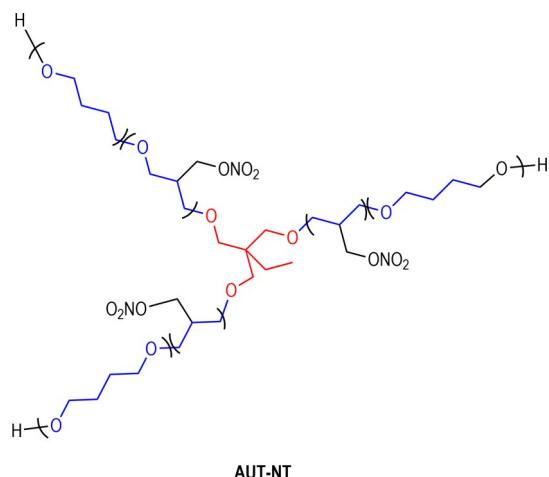
a. molecular structure of dendritic polymers^[84]b. molecular structure of star polymers^[40]

图 12 超支化聚合物分子结构

Fig.12 Molecular structures of hyperbranched polymers

型超支化叠氮聚合物,即聚-3-叠氮甲基-3-羟甲基氧烷(HBPAMHMO)(表4)。正如预期,在结构相似的情况下,HBPAMHMO($T_g=-54\text{ }^\circ\text{C}$)的 T_g 值远低于PAMMO($T_g=-44\text{ }^\circ\text{C}$)。此外,除了需要交联的线性结构外,HBPAMHMO的支链结构还具有交联活性。结果表明,在较宽的温度范围(-40~50 $\text{ }^\circ\text{C}$)下,交联HBPAMHMO的拉伸强度和断裂伸长率大于PAMMO,尤其是在低温下。

张光普等^[85]以3,3-二(氯甲基)氧杂环丁烷(BCMO)和3-乙基-3-氧杂丁环甲醇(EHO)为原料,通过调节单体BCMO、EHO混合摩尔比n,再经阳离子开环聚合和叠氮化反应制备了一系列叠氮超支化共聚物(r -POB-n)。当n=4时, r -POB-4为无定形态且工艺黏度最低。将 r -POB-4用于增塑GAP基含能热塑性弹性体推进剂,其断裂伸长率提高了约70%,稠度系数降低了约49%,黏流活化能降低了约

表4 HBPAMHMO 和 PAMMO 力学性能和安全性能^[84]Table 4 Mechanical and safety properties of HBPAMHMO and PAMMO^[84]

samples	molecular weight and distribution		T_g / °C	viscosity / Pa·s	
	M_n / g·mol ⁻¹	D			
HBPAMHMO	2830	1.12	-54	3.472 at 25 °C, 1.013 at 50°C	
PAMMO	3290	1.21	-44	3.219 at 25 °C, 0.907 at 50°C	
samples	tensile strength / MPa	elongation / %	impact sensitivity / J	friction sensitivity / J	T_g / °C
EBPAMHMO-2	8.98	277	35	324	-45
EBPAMHMO	8.22	95	25	288	-37

20%,其性能较端叠氮基聚叠氮缩水甘油醚(GAPA)增塑剂更优异,表明叠氮超支化共聚物作为增塑剂可有效提高ETPE推进剂的力学性能和工艺性能(表5)。

星状聚合物是一种特殊的超支化聚合物,其结构特征在于以一个枝化点作为中心核,从中心核呈放射状连接出三条以上线型链的共聚物^[89]。2020年,Wang等^[40]以三官能团NIMMO-THF共聚醚(T-NT)和异氰酸烯丙酯为原料,合成了一种与饱和HTPB骨架具有相似性的含三种官能团的烯丙基氨基甲酸乙酯-四氢呋喃共聚醚(AUT-NT)。研究表明,基于该黏

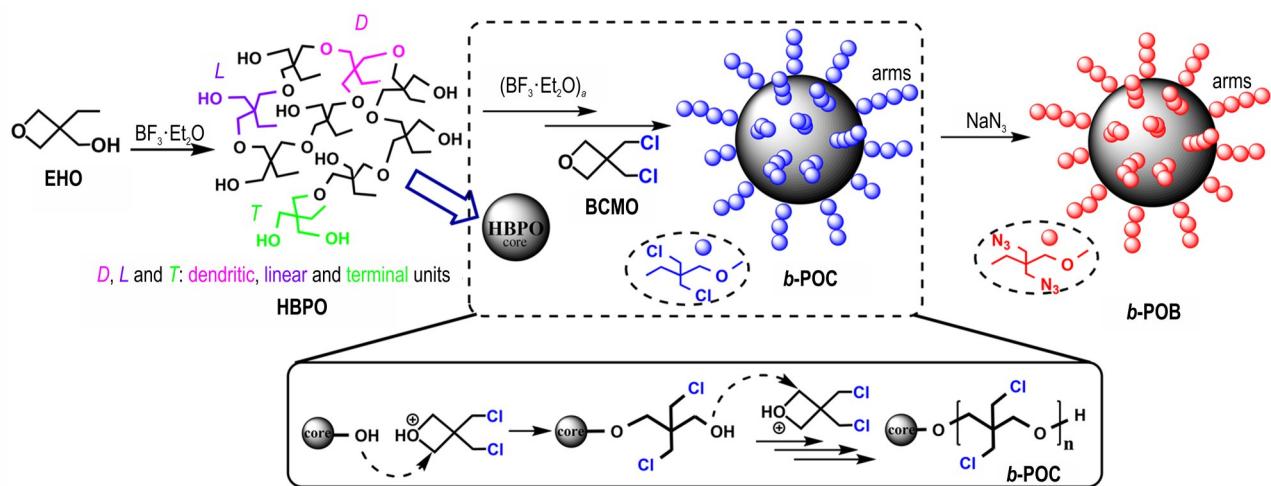
合剂体系的拉伸强度从1.0 MPa增加到了3.0 MPa,断裂伸长率从135%增至300%,体现了该黏合剂的良好力学性能。

Zhang等^[90-93]在末端羟基为起始点的超支化聚(3-乙基3-(羟甲基)氧烷)(HBPO)的基础上,通过阳离子开环共聚进一步引入BAMO和GAP环醚单体,得到超支化星形共聚物POG-*n*和POB-*n*(*n*为臂数)。*b*-POB的合成路线如图13所示。与PBAMO和GAP相比,POB-*n*和POG-*n*具有较低的 T_g 值、较低的机械刺激敏感性和更好的力学性能。综合考虑,*b*-POB-4(T_g =-38 °C)和POG-8(T_g =-51.3 °C)的 T_g 值最低。

表5 超支化聚合物的玻璃化转变温度和力学性能

Table 5 Glass transition temperatures and mechanical properties of hyperbranched polymers

samples	molecular weight and distribution		T_g / °C	viscosity / Pa·s		tensile strength / MPa	elongation/%
	M_n / g·mol ⁻¹	D		30 °C	60 °C		
r-POB-4 ^[85]	6330	1.61	-37	-	-	-	-
AUT-NT ^[40]	-	-	-	-	-	3	300
T-NT ^[40]	-	-	-	-	-	1	135
<i>b</i> -POB-4 ^[90]	5300	1.6	-38	81.2	5.8	-	-
POG-8 ^[92]	9530	1.2	-51	3.38	0.54	-	-

图13 *b*-POB 的合成^[90]Fig.13 Synthesis route of *b*-POB^[90]

超支化聚合物的合成主要是通过阳离子开环聚合的方法实现。超支化聚合物具有更高的能量密度、更低的黏度和 T_g ,从而表现出优异的流变性能。与均聚物相比,超支化叠氮聚合物的 T_g 更低,拉伸强度和断裂伸长率有所提升,整体力学性能得到增强。此外,星状超支化聚合物相比于均聚物,表现出更低的 T_g 、更低的机械刺激敏感性和更好的力学性能。

综上所述,由于柔性侧链的内增塑作用,分子间力得以增强,有效提高了含能聚醚黏合剂的力学性能和加工性能。基于含柔性侧链聚醚黏合剂的研究才刚刚兴起,柔性侧链的长度、含量、连接位点等相关机理的研究尚不明确。鉴于引入柔性侧链在增强分子链柔顺性方面的独特优势,未来有望实现更多具有优良力学特性的含能聚醚黏合剂的合成与应用。

3 总结与展望

含能聚醚黏合剂作为固体推进剂的骨架和基体,其性能对复合固体推进剂能量水平、力学性能等有着直接的影响。在含能聚合物的设计与合成研究中,高能基团的存在导致聚合物力学性能下降。因此,在保持高能量密度的同时提升其力学性能,已成为亟待解决的瓶颈问题。从20世纪80年代以来,在含能聚醚中引入柔性链段提高柔顺性已经得到学者们的广泛关注。通过共聚引入柔性链段能够提高分子链的柔顺性,有效调控力学性能,获得综合性能更优的新型含能黏合剂。基于在高能化和内增塑方面突出的优势,该方法在含能聚醚黏合剂合成与制备领域具有重要的应用潜力。本文从聚醚中主链柔性链段、侧链柔性链段两个方面详细介绍了提高聚醚柔顺性多种方法,以及相应共聚物合成与制备方面的研究进展。在主链柔性链段中,通过无规共聚、多嵌段共聚、三嵌段共聚等线性共聚的方法引入柔性较好的链段,可增强分子链的柔顺性,从而提高含能聚醚黏合剂的力学性能。同时,基于无规共聚物存在的固化效果差、难以回收、安全性低等问题,介绍了嵌段共聚物的优势和最新研究进展。在侧链柔性链段中,引入C-H类柔性侧链能够明显提高分子链的柔顺性,降低含能聚合物的 T_g 和黏度;含C-F类侧链的聚合物由于氟元素的引入,黏合剂的力学性能提高;利用点击反应制备的含三唑类侧链聚合物的拉伸强度和抗冲击性明显增强;三维超支化类聚合物表现出高能量密度、较低的黏度和 T_g ,同时具备优异的流变性能和良好的力学性能。

虽然含柔性链段含能聚醚的设计和合成已经得到很多研究者的关注,但相关研究仍存在不足,未来研究可围绕以下几个方面深入展开:

(1) 在序列调控方面,采用如点击反应的新型共聚方法,能有效控制聚合物的结构和长度,合成方法较为简单、安全,对于制备加工性能好、低 T_g 的高聚物有重要意义,但序列长度不能得到精确控制,在合成工艺上需进一步优化,合成序列结构、长度与分子量均可控的聚合物;

(2) 在嵌段共聚方面,改变含能嵌段类型和长度可以在一定程度上保留各组分单元的优点,可尝试引入与HTPE类似柔顺性更好的嵌段共聚醚作为大分子引发剂,与具有氧化性、燃烧性等功能性基团的单体共聚,实现聚合物的多功能集成,提高聚醚的嵌段单元多样性及黏合剂的综合性能;

(3) 在支链结构方面,可在现有研究基础上寻找新的侧链,探究柔性侧链的长度、含量和连接位点对分子链柔顺性的影响,提高高分子链的柔顺性,并赋予高分子链独特的性能。

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Research Progress on the Design of Flexible Chain Structures for Energetic Polyether Binders

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Abstract: Energetic polyether binders, as the backbone and matrix of propellants, are the basis to improve the energy levels, mechanical properties, and processing properties of propellants. However, the polar energetic groups in polyethers hinder the movement of molecular chains and reduce the flexibility of polymer chains, leading to a decrease in mechanical properties and restricting the development of energetic solid propellants. Copolymerization modification of energetic polyethers and introduction of flexible structural units to improve flexibility are effective methods to obtain polyethers with diverse structures and adjustable properties. This paper summarizes different types of flexible chain segments introduced into energetic polyethers in recent years, and discusses from the perspectives of main-chain flexible chain segments and side-chain flexible chain segments to explain their effects on the mechanical properties and processing performance of energetic polyether binders. Future design of the flexible chain structure of energetic polyether binders is also discussed, which will provide a direction for the design and development of new types of energetic binders.

Key words: polyether binders; polymer chain; mechanical property; solid propellant

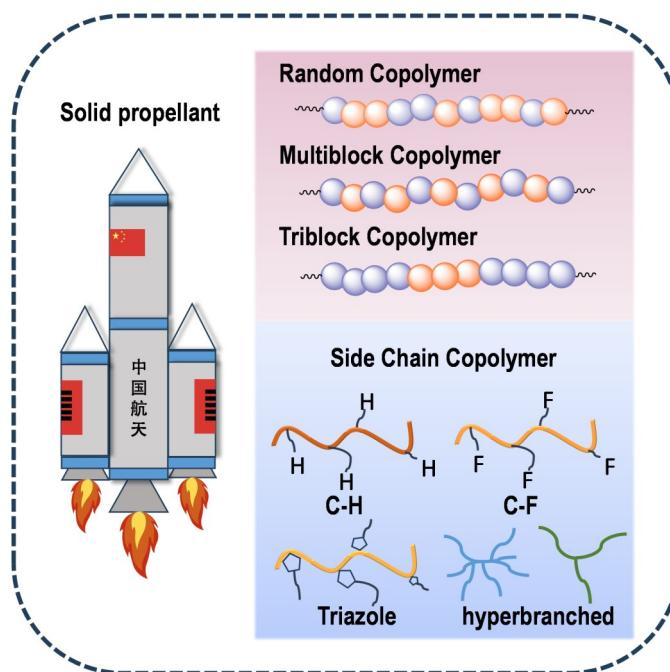
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图文摘要：



Solid propellant is an important propulsion energy source for aerospace vehicles, and energetic polyether binders are the basis of propellant. However, polar energetic groups lead to poor mechanical properties. The exploration of new polyether binders will help to improve the comprehensive performance of propellants, which is of great value for engineering applications. Copolymerization to introduce flexible chain segments to improve flexibility is an effective method to enhance the performance of polyethers. In this study, different types of flexible chain segments introduced into energetic polyethers in recent years are reviewed, and various methods to improve the flexibility of polyethers are introduced in detail from the perspectives of main-chain flexible chain segments and side-chain flexible chain segments, as well as the progress of the synthesis and preparation of the corresponding copolymers. Future directions for the structural design of the flexible chain structure of the energetic polyether adhesives is discussed, which can provide a certain reference to the development of the new energetic adhesives.